

Parallel three-dimensional simulations of quasi-static elastoplastic solids

Nicholas M. Boffi & Chris Rycroft

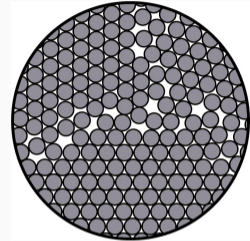
July 16, 2019

Harvard University, Department of Applied Mathematics.
CSGF Program Review 2019.

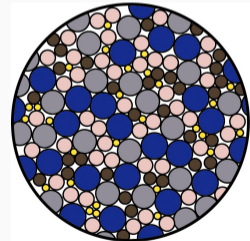
Bulk metallic glasses (BMGs)

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- Solid material held together by metallic bonds with an amorphous atomic-scale structure.



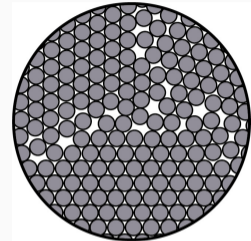
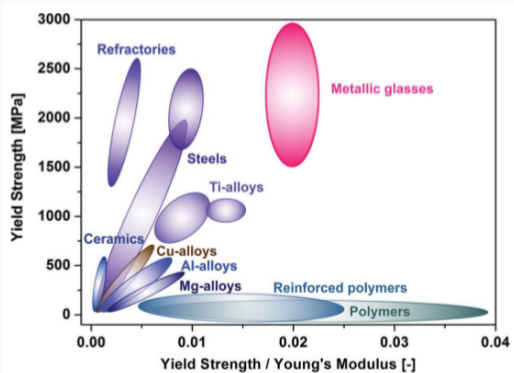
Crystalline structure



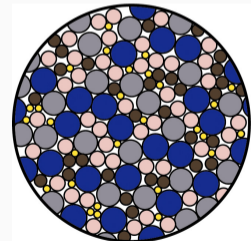
Amorphous structure

Bulk metallic glasses (BMGs)

- Solid material held together by metallic bonds with an amorphous atomic-scale structure.
- Metallic bonds lead to **dense packing** unlike oxide glasses.
 - Stronger than steel, with high strength to elasticity ratio.
 - Can be processed and molded like a plastic.



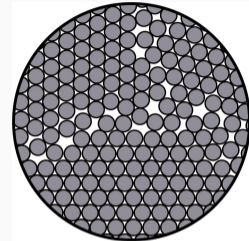
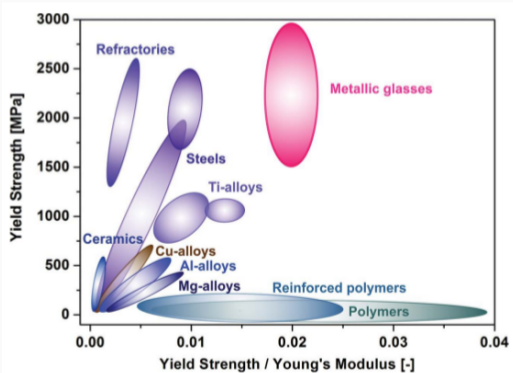
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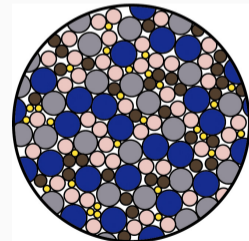
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- Metallic bonds lead to **dense packing** unlike oxide glasses.
 - Stronger than steel, with high strength to elasticity ratio.
 - Can be processed and molded like a plastic.
- Applications limited by catastrophic failure mechanism: **shear banding**.



Crystalline structure



Amorphous structure

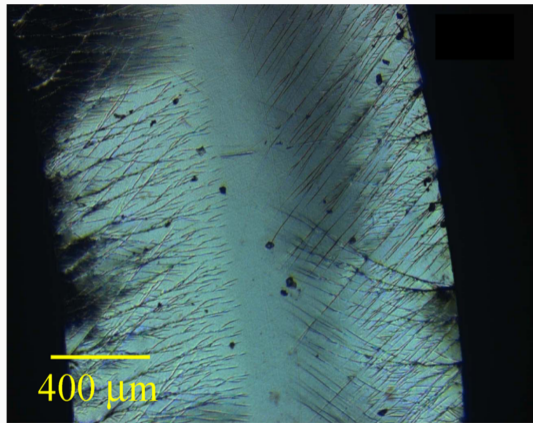
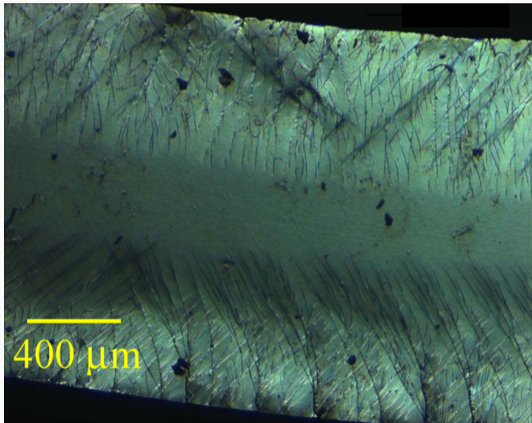
Images taken from Schroers & Johnson, *Phys. Rev. Let.*, 2004, 93, 255506

Shear bands

- **Shear band**: rapid (ms) narrow (10 nm) localization of plastic deformation due to a strain-softening instability.

Shear bands

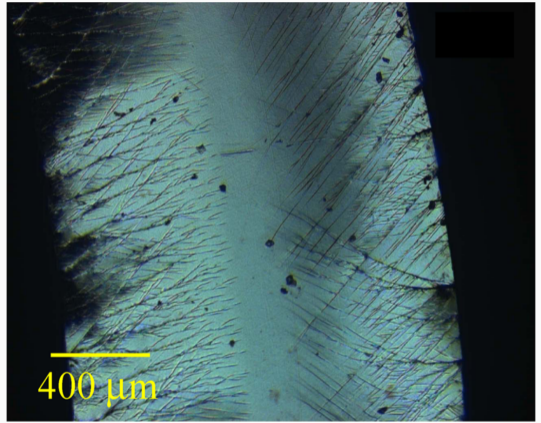
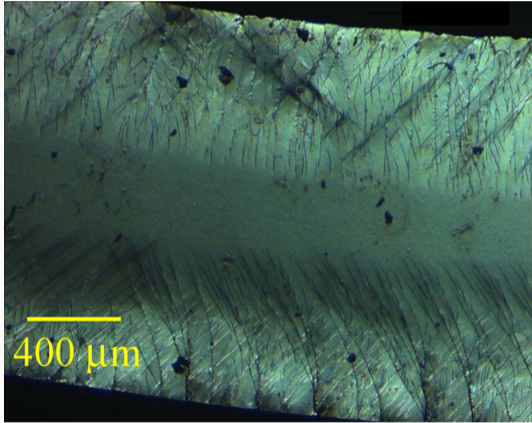
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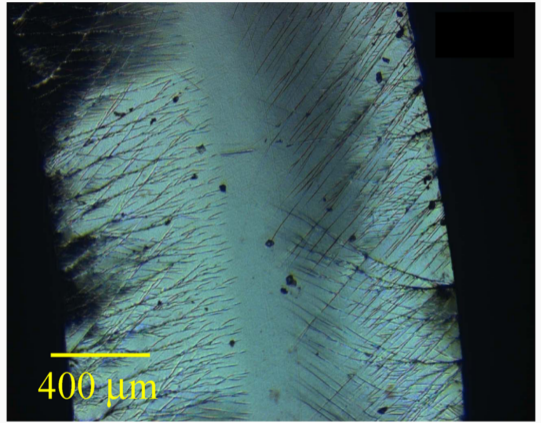
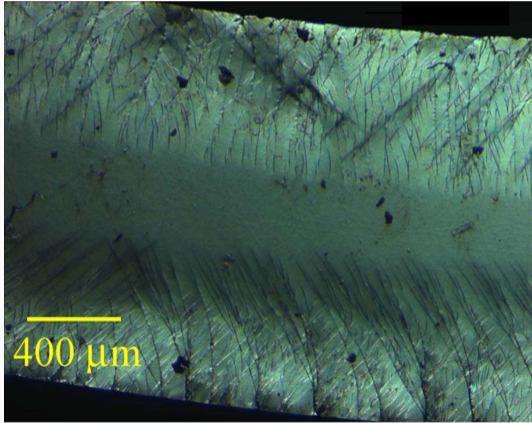
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- Physics: Any theory of amorphous plasticity must be able to predict structure and formation of shear bands.



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Shear bands

- **Shear band**: rapid (ms) narrow (10 nm) localization of plastic deformation due to a strain-softening instability.
- Physics: Any theory of amorphous plasticity must be able to predict structure and formation of shear bands.
- Applications: Until shear bands are understood and controlled, BMGs cannot be used in practice.

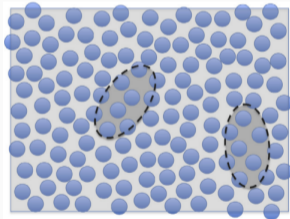


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- **STZs**: localized regions likely to undergo small-scale plastic rearrangements.

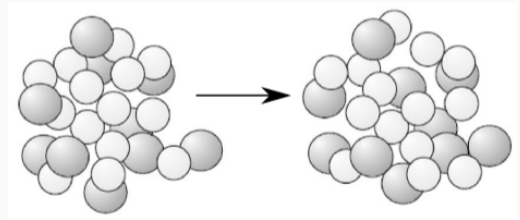
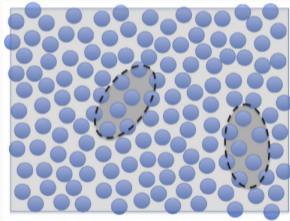
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Shear transformation zone theory of amorphous plasticity

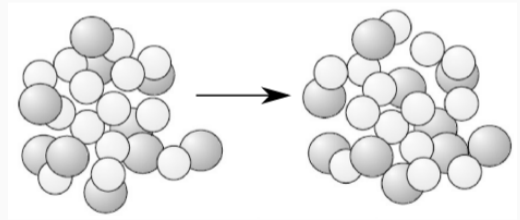
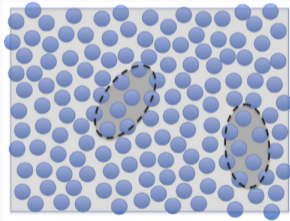
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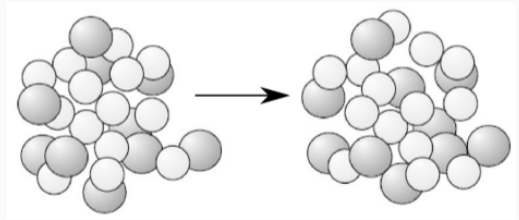
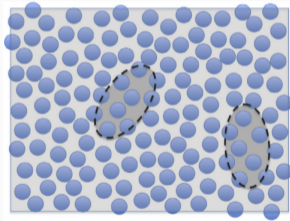
- **STZs**: localized regions likely to undergo small-scale plastic rearrangements.
- Randomly distributed with Boltzmann density in an effective disorder temperature χ .



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Shear transformation zone theory of amorphous plasticity

- **STZs**: localized regions likely to undergo small-scale plastic rearrangements.
- Randomly distributed with Boltzmann density in an effective disorder temperature χ .
- χ is measured in Kelvin, $\chi \propto \frac{\partial(\text{Config. Energy})}{\partial(\text{Config. Entropy})}$, but distinct from usual kinetic/vibrational temperature T .



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Continuum theory

Linear Elasticity

Shear Transformation Zone Theory

Linear Elasticity

$$\underbrace{\frac{\mathcal{D}\sigma}{\mathcal{D}t}}_{\text{Truesdell Rate}} = \overset{\text{Stiffness}}{\hat{\mathbf{C}}} : \underbrace{\mathbf{D}}_{\text{Elastic Part}}^{\text{el}}$$

Shear Transformation Zone Theory

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Stiffness

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$$\underbrace{\rho \frac{d\mathbf{u}}{dt}}_{m \times a} = \underbrace{\nabla \cdot \sigma}_{\text{Net force}}$$

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Shear Transformation Zone Theory

$$\underbrace{\mathbf{D}^{\text{pl}}}_{\text{Plastic rate}} = \underbrace{D^{\text{pl}} f(\sigma)}_{\propto \text{deviatoric}}$$

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$$\underbrace{\mathbf{D}^{pl}}_{\text{Plastic rate}} = \underbrace{D^{pl} f(\sigma)}_{\propto \text{deviatoric}}$$

$$\underbrace{\frac{d\chi}{dt}}_{\text{Advective rate}} = \frac{\overbrace{D^{pl} \bar{s}}^{\text{Relaxation term}}}{s_y c_0} (\chi_\infty - \chi) + \underbrace{l^2 \nabla \cdot (D^{pl} \nabla \chi)}_{\text{Diffusive term}}$$

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hypo-elastoplastic *assumption*

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$D^{pl} \propto$ (Density of STZs) \times (Arrhenius barrier crossing)

\times (Rates for forward and reverse transitions) $\times \left(1 - \frac{s_y}{\bar{s}}\right)$

$\times \begin{cases} 1 & \text{if local stress above yield stress} \\ 0 & \text{else} \end{cases}$

The quasi-static and incompressible limits

Hypo-elastoplastic long-time limit

Incompressible limit of Navier-Stokes

From Navier-Stokes to hypo-elastoplasticity

The quasi-static and incompressible limits

Hypo-elastoplastic long-time limit

$$\underbrace{\frac{\mathcal{D}\sigma}{\mathcal{D}t}}_{\text{hypo-elastoplastic equation}} = \mathbf{C} : (\mathbf{D} - \mathbf{D}^{\text{pl}})$$

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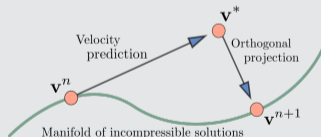
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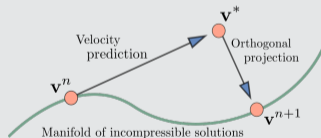
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$$\sigma_{\text{H.E.P.}} \iff \mathbf{v}_{\text{N.S.}}$$

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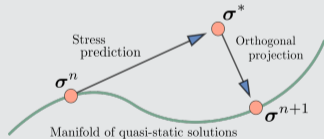
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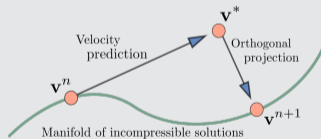
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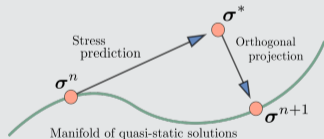
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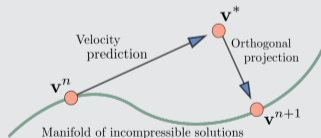
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- Any algorithm for Navier-Stokes should work for hypo-elastoplasticity.
- This analogy is **independent** of the plasticity model.

A quasi-static projection method

Hypo-elastic equation

- Continuous-time:

$$\frac{\mathcal{D}\sigma}{\mathcal{D}t} = \mathbf{C} : (\mathbf{D} - \mathbf{D}^{\text{pl}}).$$

- Semi-implicit Euler discretization:

$$\frac{\sigma^{n+1} - \sigma^n}{\Delta t} = \left(\frac{D\sigma}{Dt} \right)^n - (\mathbf{C} : \mathbf{D}^{\text{pl}})^n + (\mathbf{C} : \mathbf{D})^{n+1}$$

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Advection step

- Just drop the $\mathbf{C} : \mathbf{D}$ term!

$$\frac{\sigma^* - \sigma^n}{\Delta t} = \left(\frac{D\sigma}{Dt} \right)^n - (\mathbf{C} : \mathbf{D}^{\text{pl}})^n$$

- Parallelized using domain decomposition and MPI (along with χ update).
- Solved on a staggered grid with σ and χ at cell centers and \mathbf{u} at cell corners.

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Projection step

- σ^{n+1} can be computed from σ^* just by adding back in the $\mathbf{C} : \mathbf{D}$ term.

$$\frac{\sigma^{n+1} - \sigma^*}{\Delta t} = \mathbf{C} : \mathbf{D}^{n+1}$$

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Linear system for the velocity

- Take divergence of the projection step:

$$\nabla \cdot \sigma^* = -\Delta t \nabla \cdot (\mathbf{C} : \mathbf{D}^{n+1})$$

- Solved using a custom C++ implementation of the geometric multigrid method.
- Implementation is parallelized using MPI and interfaces with the advection step.
- Optimal MPI decomposition computed at each level.
- C++ templates to solve for an arbitrary datatype at each point in space - in this case, a 3-vector.

Molecular Dynamics

Continuum-Scale

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- Physically exact up to integration errors and choice of atomic potentials.

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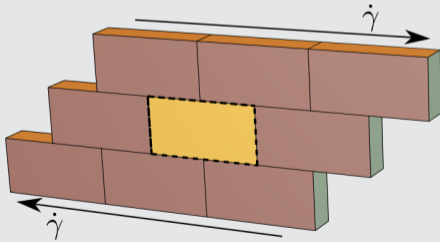
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- Fast, scalable simulation.

Simulation methods for BMGs

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- Computationally expensive!
- Typically employs Lees-Edwards boundary conditions.



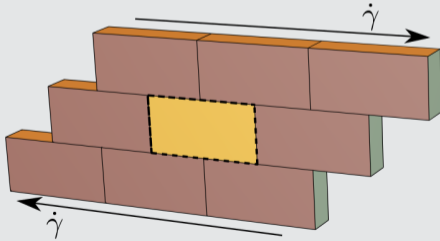
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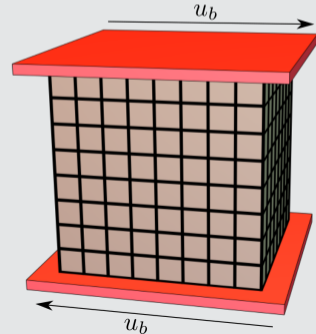
Molecular Dynamics

- Physically exact up to integration errors and choice of atomic potentials.
- Complete information about atomic structure and atomic configurations.
- Computationally expensive!
- Typically employs Lees-Edwards boundary conditions.



Continuum-Scale

- Requires constitutive equations and a phenomenological plasticity model.
- Fine-scale details have been coarse-grained into internal model variables.
- Fast, scalable simulation.
- Typically employs parallel-plate boundary conditions.

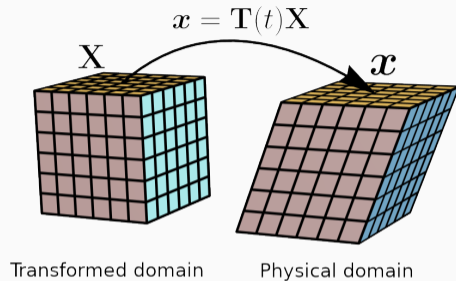


Transformation methodology

- Reference or *transformed* domain with coordinate \mathbf{X} .
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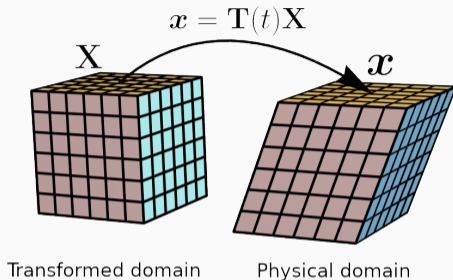


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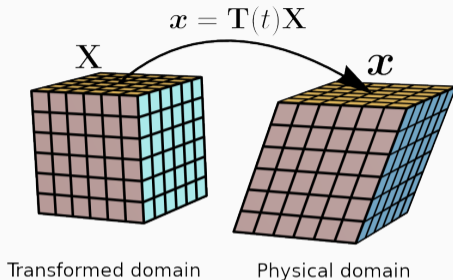


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Transformed Evolution Equations

$$\frac{\partial \Sigma}{\partial t} = -(\mathbf{V} \cdot \nabla_{\mathbf{X}}) \Sigma - \text{tr}(\mathbf{L}) \Sigma + \Sigma (\mathbf{T}^T \mathbf{L}^T \mathbf{T}^{-T}) + (\mathbf{T}^{-1} \mathbf{L} \mathbf{T}) \Sigma$$

$$+ \mathbf{T}^{-1} \left(\mathbf{C} : (\mathbf{D} - \mathbf{D}^{pl}) - \frac{\partial \mathbf{T}}{\partial t} \Sigma \mathbf{T}^T - \mathbf{T} \Sigma \frac{\partial \mathbf{T}^T}{\partial t} \right) \mathbf{T}^{-T}.$$

$$\frac{\partial \mathbf{V}}{\partial t} = -(\mathbf{V} \cdot \nabla_{\mathbf{X}}) \mathbf{V} + \frac{\partial \mathbf{T}^{-1}}{\partial t} \mathbf{T} \mathbf{V} + \mathbf{T}^{-1} \left(\mathbf{T}^{-T} \nabla_{\mathbf{X}} \cdot (\mathbf{T} \Sigma \mathbf{T}^T) - \frac{\partial^2 \mathbf{T}}{\partial t^2} \mathbf{X} - \frac{\partial \mathbf{T}}{\partial t} \mathbf{V} \right).$$

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Quasi-static shear simulations

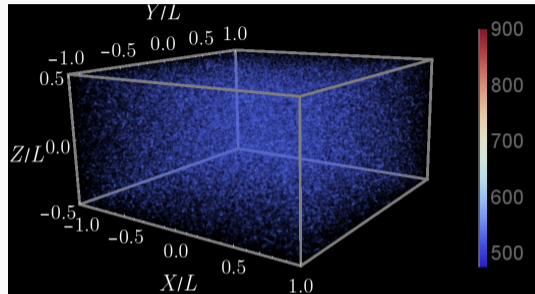
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- $\approx 67,000,000$ grid points in the bulk.
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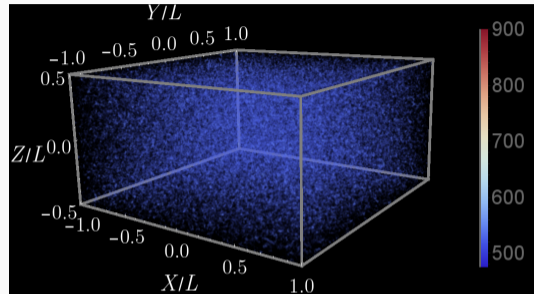


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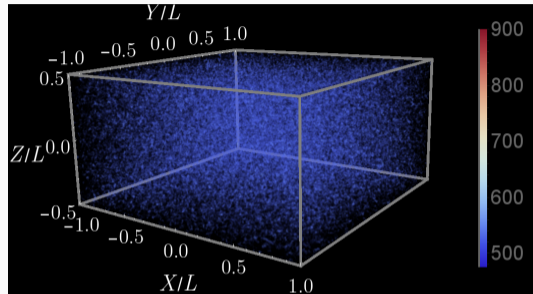


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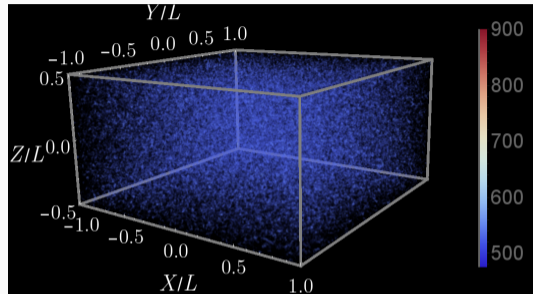
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- ≈ 4 day simulation time with 32 threads.



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Conclusions, Future Directions, and Acknowledgments

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